Synthesis of flower-like CuO nanostructures as a sensitive sensor for catalysis

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ABSTRACT

The flower-like CuO nanostructures were hydrothermally synthesized without using any template. The influences of hydrothermal temperature and time on the growth of nanostructures were investigated. The samples were characterized by means of scanning electron microscope (SEM), X-ray powder diffraction (XRD), transmission electron microscope (TEM), high-resolution transmission electron microscope (HRTEM), selected area electron diffraction (ED), and N2 adsorption isotherm. Interestingly, these architectures are made of three-order structures. The formation mechanism of the flower-like CuO was proposed and explained. Furthermore, the chemiluminescence (CL) and catalysis properties of the flower-like CuO were also investigated. The flower-like nanostructures showed the high-CL intensities and reactive activities for CO oxidation. The flower-like CuO can be used to fabricate a highly sensitive CL detector. This CL mode is a rapid and effective method for the selection of new catalysts from thousands of materials.

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1. Introduction

Recently, the research on the shape control of various nanostructures has been widely developed because of their morphology dependent properties. The cuprous oxide (Cu2O) and copper oxide (CuO) with different morphologies have also been prepared by several routes, such as thermal oxidation [1], simple solution [2], simple hydrolysis [3], template-based sol–gel [4], and electrochemical [5]. Besides, the hydrothermal [6–10], solvothermal [11,12] and microwave-hydrothermal [13,14] methods have also been employed to control the morphologies of these oxides. Furthermore, it has been shown that the addition of different surfactants can effectively promote the formation of CuO/Cu2O nanostructures (2D and 3D) [15–18].

CuO is a p-type semiconductor with a narrow bandgap (1.2 eV), which shows the interesting electrochemical and catalytic properties. CuO has received considerable attention due to its potential applications in many fields, such as catalysis, gas sensors, and superconductors [19–21]. The advanced structures of CuO with varied shapes have been obtained, such as nanowires, nanoribbons, hollow, dendrites, dendelions, prickly, flowers and film [22–32]. Liu and Zeng [27] have reported the self-organization of CuO hierarchical microspheres. The microspheres are similar to the spherical assemblies or “dandelions” with a puffy appearance. Xu et al. [28] have also prepared CuO prickly microspheres using a simple solution method. Besides, CuO whisker assemblies have also been synthesized by a microwave-induced process [33]. CuO is an excellent catalyst for CO oxidation, which has been widely used in indoor air cleaning, fuel cells and automotive exhaust treatment [34–36]. It is important to explore oxide-based hierarchical structures for the applications in catalysis and nanodevices. To the best of our knowledge, however, the majorities of researches mainly focus on the synthesis of these advanced micro-/nanostructures; the physico-chemistry properties of these advanced structures have been scarcely reported.

In this work, the flower-like CuO nanostructure was hydrothermally synthesized using copper threads as precursor. The influences of hydrothermal temperature and time on the growth of crystals were investigated, and the formation mechanism of CuO nanostructure was proposed and discussed. The chemiluminescence (CL) and catalysis properties of the samples were investigated. Furthermore, a highly sensitive CL detector was fabricated using these nanostructures.
Table 1

Preparation, surface areas, crystal structures, CL intensities, and catalytic activities of the samples

<table>
<thead>
<tr>
<th>Sample</th>
<th>$T$ (°C), time (h)</th>
<th>$S_{BET}$ (m$^2$ g$^{-1}$)</th>
<th>CL intensity ($\times 10^{4}$ a.u.)</th>
<th>$T_{50}$ (°C)$^b$</th>
<th>Crystal structure</th>
</tr>
</thead>
<tbody>
<tr>
<td>S1</td>
<td>100, 24</td>
<td>–</td>
<td>–</td>
<td>–</td>
<td>Cu(OH)$_2$</td>
</tr>
<tr>
<td>S2</td>
<td>120, 24</td>
<td>15.5</td>
<td>1.6</td>
<td>160</td>
<td>CuO</td>
</tr>
<tr>
<td>S3</td>
<td>140, 2</td>
<td>–</td>
<td>–</td>
<td>–</td>
<td>Cu(OH)$_2$</td>
</tr>
<tr>
<td>S4</td>
<td>140, 8</td>
<td>14.5</td>
<td>2.0</td>
<td>155</td>
<td>CuO</td>
</tr>
<tr>
<td>S5</td>
<td>140, 12</td>
<td>10.3</td>
<td>2.5</td>
<td>150</td>
<td>CuO</td>
</tr>
<tr>
<td>S6</td>
<td>140, 24</td>
<td>9.7</td>
<td>2.8</td>
<td>145</td>
<td>CuO</td>
</tr>
<tr>
<td>S7</td>
<td>160, 24</td>
<td>9.2</td>
<td>3.1</td>
<td>140</td>
<td>CuO</td>
</tr>
<tr>
<td>S8</td>
<td>180, 24</td>
<td>8.4</td>
<td>3.7</td>
<td>130</td>
<td>CuO + Cu$_2$O</td>
</tr>
</tbody>
</table>

$^a$ $S_{BET}$, surface area calculated by the BET method.

$^b$ $T_{50}$, temperature at 50% CO conversion.

2. Experimental procedure

2.1. Synthesis

In the experiments, all chemicals were analytic grade and purchased from Beijing Chemicals Company of China. They were used without further purification.

Typically, the copper threads (99.99%, 10 mm $\times$ 1 mm) were immersed in a 2.0 M HCl solution under ultrasonic stirring; after 10 min, the copper threads were washed with acetone and deionized water in sequence to remove the surface impurities. 1.0 g of the fresh copper threads was put into a 50-mL Teflon-lined autoclave containing 40 mL of 0.5 mol L$^{-1}$ K$_2$Cr$_2$O$_7$ and 5 mL of 98 wt.% H$_2$SO$_4$ solutions, while the pH value of the solution was kept at 2–3. The autoclave was then sealed and heated to 140 °C in an oven.

Fig. 1. SEM micrographs of the samples prepared at different reaction temperatures for 24 h: (a) $T$ = 100 °C, (b) $T$ = 120 °C, (c) $T$ = 140 °C, and (d) $T$ = 160 °C.
After 24 h, the remained copper threads were taken out from the mixture. The solids were separated by centrifugation, washed with deionized water, and dried at 80 °C overnight. In order to explore the formation of the nanostructures, the hydrothermal temperature and time were varied, respectively.

2.2. Characterization

The morphologies of the samples were characterized by a scanning electron microscope (SEM, KYKY 2800) equipped with a Link Isis EDS analyzer. The acceleration voltage was 15 kV and the current was 1.2 nA. The morphologies of the samples were also observed on a transmission electron microscope (TEM, JEOL 2000CX) with an accelerating voltage of 200 kV. High-resolution transmission electron microscope (HRTEM, JEOL JEM-2010) was used to characterize the surface structures of the samples. The powders were ultrasonically dispersed in ethanol, and then deposited on a thin amorphous carbon film supported by a copper grid. The crystal structures of the samples were characterized by X-ray powder diffractometer (XRD, Rigaku D/MAX-RB), using graphite monochromatized Cu Kα radiation (λ = 0.154 nm), operating at 40 kV and 50 mA. The XRD patterns were obtained in the range of 10°–70° (2θ) at a scanning rate of 5°·min⁻¹. A nitrogen adsorption isotherm was performed at −196 °C on a Micromeritics ASAP2010 gas adsorption analyzer. The sample was degassed at 200 °C for 3 h before the measurement. The surface area was calculated by the Brunauer–Emmett–Teller BET method [37].

2.3. Evaluation of chemiluminescence and catalytic properties

CL properties were evaluated on a CL detection system, which is made by Biophysics Institute of Chinese Academy of Science in China. The powders were sintered to form a 0.2-mm thickness coating on the heating tube. The sensor was put into a quartz tube with the inner diameter of 12 mm. The mixed gases of air and CO flowed through the quartz tube at a rate of 100 mL·min⁻¹, and CO concentration in the mixed gases was kept at 400 ppm. The bandpass filter of 425-nm band length was used. The resulting CL signal at 200 °C was directly measured with a BPCL ultra-weak luminescence analyzer.

The oxidation reaction of CO was carried out in a conventional flow system at an atmospheric pressure. 0.1 g of the catalyst was loaded in a quartz reactor (inner diameter: 5 mm), with quartz beads packed at both ends of the catalyst bed. The thermal couple was inserted in the catalyst bed to monitor the reaction temperature since CO oxidation is an exothermic reaction. Before each run, the catalyst bed was flushed with air (100 mL·min⁻¹) at 300 °C for 1 h, so as to remove the adsorbed species from the catalyst surface, and then cooled to 30 °C. The mixed gases of 2 vol.% CO and 98 vol.% air were fed to the catalyst bed at a certain flowing rate of 100 mL·min⁻¹. The inlet and outlet gas compositions were analyzed by an on-line gas chromatograph with a GDX-403 GC-column (1.5 m × 4 mm) and a hydrogen flame ionization detector (FID).

3. Results and discussion

3.1. Formation and characterization of the nanostructures

Table 1 gives the preparation conditions of the samples. In order to investigate the influence of hydrothermal temperature on the nanostructures, the samples were prepared at 100, 120, 140, 160, and 180 °C for 24 h, respectively. Their typical SEM micrographs and XRD patterns are shown in Figs. 1 and 2, respectively. At 100 °C, a large number of the nanoparticles were prepared (Fig. 1(a)). All the diffraction peaks of the sample could be indexed to an orthorhombic structure of Cu(OH)₂ with the lattice constants a = 2.951 Å, b = 10.59 Å and c = 5.273 Å (JCPDS 35-0505) (Fig. 2(a)). This is also confirmed by the color of the product (the characteristic blue color of Cu(OH)₂). This indicates that CuO crystals cannot be formed due to the low temperature. When the temperature was increased to 120 °C, both small flowers and nanoparticles were obtained (Fig. 1(b)). At 140 and 160 °C, a large number of the well-organized flower-like structures were obtained (Fig. 1(c) and (d)). Interestingly, many of crystal ornaments can be clearly observed on the petals. All the diffraction peaks of the samples can be indexed to a monoclinic structure of CuO with the lattice constants a = 4.6837 Å, b = 3.4226 Å and c = 5.1288 Å (JCPDS 45-0937) (Fig. 2(b)–(d)). At 180 °C, the flower-like structures have not been prepared, but the irregular particles formed (not showing the SEM micrographs). Fig. 2(e) shows that this sample consists of CuO and a small amount of Cu₂O. The formation of Cu₂O can be also confirmed by the reddish color of the product. This result indicates that Cu(II) can be reduced to Cu(I) at a high-hydrothermal temperature. The result is well consistent with the reports by other researchers [38,39]. It is obvious that the hydrothermal temperature has a significant influence on the morphology and structure of the product. In this study, the well-organized CuO flower-like nanostructures could form at 140–160 °C.

To explore the formation of the nanostructures, the hydrothermal time was varied from 2, 8, 12, to 24 h while the hydrothermal temperature was kept at 140 °C. After 2 h, a large number of nanoparticles were formed (Fig. 3(a)). When the reaction time was prolonged to 8 h, the small flowers formed, and the lengths of the petals are about 1.5–2 μm (Fig. 3(b)). After 12 h, the well-organized flower-like architectures formed, and the lengths of the petals reached 2.5–3 μm (Fig. 3(c)). After 24 h, the sample has a three-order structure, and the lengths of the petals were about 4–5 μm (Fig. 3(d)). The crystal ornaments can be clearly observed on the petals. Further prolonging the reaction time, the sizes of the flowers did not increase significantly. Fig. 4 shows the XRD patterns of the as-synthesized samples at 140 °C. After 2 h, the sample shows the structure of Cu(OH)₂ (Fig. 4(a)). When the reaction time was prolonged to 8, 12, and 24 h, all the samples showed the crystal structure of CuO (Fig. 4(b)–(d)). No Cu₃O₂ or Cu(OH)₂ could be detected by XRD. It is obvious that the reaction time has a significant influence on the morphology and structure of the products. It seems that the flower-like structures may grow from the nanoparticles.
Further, the sample (in Fig. 3(d)) was characterized by TEM, HRTEM and ED. Fig. 5(a) and (b) show the typical TEM micrographs of the flowers with several petals. Fig. 5(c) shows the typical TEM image of a petal. It reveals that the flower-like structure is made of the three-order structure: the nanocrystals, the petals, and the assembly of the petals. Fig. 5(d) shows the typical HRTEM image of the ornament (indicated by the arrow in Fig. 5(c)). The interplanar spacings were estimated to be 0.215, 0.232, and 0.274 nm, which correspond to the [0 0 2], [1 1 1], and [1 1 0] planes of a monoclinic CuO crystal, respectively. Fig. 5(e) shows the ED pattern of the ornament (indicated by the arrow in Fig. 5(c)). The diffraction rings could be also indexed to a monoclinic CuO structure, rather than a cubic CuO structure. The diffraction rings confirm the polycrystalline properties of the sample.

Based on the observations, the oxidation-dehydration processes may have occurred as below:

\[
3\text{Cu} + \text{Cr}_2\text{O}_7^{2-} + 8\text{H}^+ \rightarrow 3\text{Cu(OH)}_2 + 2\text{Cr}^{3+} + \text{H}_2\text{O} \quad (1)
\]

\[
\text{Cu(OH)}_2 \rightarrow \text{CuO} + \text{H}_2\text{O} \quad (2)
\]

The \text{Cr}_2\text{O}_7^{2-} ions have the high-oxidation potential in the slightly acidic solution, the redox reaction could occur between \text{Cr}_2\text{O}_7^{2-} and Cu. The \text{Cr}_2\text{O}_7^{2-} ions moved and attacked the surfaces of Cu threads, and Cu (0) was oxidized to form Cu(II) ion; at the same time, the pH value of the solution would increase slightly, resulting in the formation of \text{Cu(OH)}_2 nanocrystals. Under the hydrothermal conditions, \text{Cu(OH)}_2 could be transformed into CuO through a dehydration/dehydroxylation process. The reaction
Fig. 4. XRD patterns of the samples prepared at 140 °C for different reaction intervals: (a) t = 2 h, (b) t = 8 h, (c) t = 12 h, and (d) t = 24 h.

process can be also confirmed by the surface variation of Cu thread before and after reaction (Fig. 6). Before reaction, the surface of Cu thread was glazed; after reaction, however, Cu thread was eroded and the surface became concave. At 140 °C, the variation of the petal size with hydrothermal time was followed by SEM (Fig. 7). Before 4 h, the petal size increased slowly, indicating that the nucleation of the crystals is significant; in the range of 5–16 h, the petal size increased quickly, indicating that the growth of crystals is significant; after 16 h, the petal size did not increase significantly. This may imply that the petals likely may result from the growth of the primary nanocrystals. In the process, the small crystals grew into the petals through an Oswald ripening process. Since Cu threads were used as the precursor, we could picture that Cu solid may also provide the specific heterogeneous nucleation sites and initiate the growth of the crystals, in which the flower-like structures may have formed via a solid-solution-solid process [40–42]. With the proceeding of the reaction, the pH value of the solution increased slightly, which in turn promoted the formation of Cu(OH)2 or CuO crystals. Cu thread, providing with the specific sites, facilitated the nucleation and growth of Cu(OH)2 or CuO crystals; CuO crystals further grew into the petals; thereafter, the petals may fuse “head-to-head” into the flower-like structures while Cu threads were completely oxidized. Similar to the report by Zeng and co-workers [43], the needlelike branches can fuse side-by-side into each other.

Hydrothermal reaction is an effective method to obtain the desirable crystals, such as mild conditions, controllable morphology, low aggregation and high crystallinity. It could be assumed that the formation of the flower-like structure was controlled not only by the growth thermodynamics, but also by the growth kinetics. The formation process could be purposely divided into several processes: (i) formation of the primary nanocrystals, (ii) growth of the secondary structure and (iii) formation of the three-order structure. The growth mechanism is similar to those of the branched ZnO [43] and Cu2O crystals [44]. Fig. 8 was used to illustrate the formation of the flower-like structures. While Cu was oxidized by K2Cr2O7, Cu(OH)2 or CuO nanocrystals formed and adsorbed on the surface of Cu thread (Fig. 8(a)); these nanocrystals grew into the pristine petals through the dissolution/crystallization process, and the small petals further grew into the large ones (Fig. 8(b)). At the same time, the nanocrystals grew on these petals and formed the small ornaments (Fig. 8(c)). It seems that the nucleation and growth of the crystals could be promoted by Cu solids. While Cu threads were

Fig. 5. TEM, HRTEM micrographs and ED pattern of the CuO flowers prepared at 140 °C for 24 h: (a–c) TEM, (d) HRTEM, and (e) ED.
Fig. 6. SEM micrographs of the Cu thread before (a) and after (b) reaction at 140 °C for 2 h.

Fig. 7. The petal sizes of the CuO flowers prepared at 140 °C for different intervals: the petal sizes were obtained by SEM.

Fig. 8. The formation sketch diagram of the flower-like structures: (a) formation of nanocrystals, (b) growth of the petals, and (c) growth of small ornaments formation of the flower structures.

oxidized completely, the petals may fuse “head-to-head” to form the flower-like structures. Note that the formation of the nanostructures and the phase transformation of Cu(OH)2 into CuO have occurred simultaneously. To clearly picture what truly governs the site-specific nucleation and growth as such, our extensive studies including interface spectroscopic analysis and computational simulation are ongoing currently.

3.2. Chemiluminescence (CL) and catalysis properties of the nanostructures

Since CuO is an excellent oxidation catalyst for CO oxidation, the CL and catalysis properties of the flower-like CuO nanostructures were studied. Fig. 9 shows the CL spectra of CO oxidation over the catalysts. Herein, the catalysts synthesized at 120/24, 140/8, 140/12, 140/24, 160/24, and 180/24 (°C/h) were designated as S2, S4, S5, S6, S7, and S8, respectively. It could be observed that the CL intensities of the catalysts follow the orders: S2 (1.6 × 10^4 a.u.) < S4 (2.0 × 10^4 a.u.) < S5 (2.5 × 10^4 a.u.) < S6 (2.8 × 10^4 a.u.) < S7 (3.1 × 10^4 a.u.) < S8 (3.7 × 10^4 a.u.). This indicates that the different amounts of the CO₂ molecules may be
produced. It is generally accepted that the CL reaction can be represented with the following equations \[45,46\]:

\[
\text{CO(ads)} + \text{O} \rightarrow \text{CO}_2(\text{ads}) + \text{Energy} \rightarrow \text{CO}_2^*(\text{ads}) \rightarrow \text{CO}_2 + h\nu
\]

While the CO molecules were oxidized on the solid surface, an amount of energy was released, which would be absorbed by the produced \(\text{CO}_2\) molecules. As a result, the \(\text{CO}_2\) molecules would jump from the electronic ground state to the electronic excited state \((\text{CO}_2^*)\). While the electronic excited \(\text{CO}_2^*\) molecules decayed to the electronic ground state, a weak light would be emitted. The CL intensity is linearly proportional to the produced \(\text{CO}_2\) concentration \[46\]. The CL intensity may closely correlate with the catalytic reaction.

Further, the reaction activities of CO oxidation over these catalysts were also evaluated. Their activities are shown in Fig. 10 and summarized in Table 1. The temperatures at 50% CO conversion \(\left(T_{50}\right)\) over S2, S4, S5, S6, S7, and S8 are 160, 155, 150, 145, 140, and 130 \(^\circ\)C, respectively. The XRD results showed that the crystallinities of the catalysts follow the orders: S2 < S4 < S5 < S6 < S7 < S8; however, the BET areas of the catalysts follow the orders: S2 (15.5 m\(^2\) g\(^{-1}\)) > S4 (14.5 m\(^2\) g\(^{-1}\)) > S5 (10.3 m\(^2\) g\(^{-1}\)) > S6 (9.7 m\(^2\) g\(^{-1}\)) > S7 (9.2 m\(^2\) g\(^{-1}\)) > S8 (8.4 m\(^2\) g\(^{-1}\)). The results mean that the crystallinity of the catalyst played the dominating role in the catalytic oxidation of CO, but the BET area played the minor role. The amounts of amorphous form present in the low-crystallinity catalyst may not favor for the oxidation of CO. As a result, the high-crystallinity catalyst showed a high-catalytic activity. The highest activity of the S8 sample could be ascribed to the high crystallinity and the presence of Cu2O. It has been reported that Cu2O had a higher activity of CO oxidation than CuO crystal \[47\]. It is most important that the order of the CL intensities of the catalysts is well consistent to that of their reaction activities. The CL spectra could be used to quantitatively evaluate a given catalytic reaction.

Based on the good correlation between CL intensity and activity, we have developed a CL-based detector with the multi-regions (Fig. 11). This technique could be potentially applied to explore the new catalysts. The different catalysts can be coated on the different regions, and the CL signals can be acquired by adjusting the location of the coating. The multi-region CL detector is simple and rapid for screening the catalysts. It is known that the catalyst evaluation is generally performed by testing the reaction activity, in which the expensive equipment is needed and the test time is long; therefore, the conventional selection mode is expensive, time-consuming and laborious. For the CL mode, no complicated and expensive equipment is needed and the CL spectra can be determined within a few minutes; therefore, the CL mode is facile, rapid, and low cost. It should be noted that the present CL method is effective for judging activity, while possibly insufficient for judging selectivity, which needs further research.

4. Conclusions

The flower-like CuO nanostructures can be hydrothermally synthesized at 120–160 \(^\circ\)C for 12–24 h. A highly sensitive CL sensor could be fabricated using the flower-like nanostructures, basing on the close correlation of CL intensity with reaction activity of the catalyst. This CL mode could be facile and effective for screening the excellent catalyst from thousands of materials. It is important to explore oxide-based hierarchical structures for applications in nanodevices.

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